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## *Communications*

## **A Liquid MOCVD Precursor for Thin Films of CdO**

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Many of the electronic and optical properties<sup>1</sup> of  $CdO$ are originated by its nonstoichiometric composition which, in turn, strongly depends on the synthetic procedure adopted. In fact, the presence of cadmium interstitials,  $Cd^+$  ions, or oxygen vacancies gives rise to donor states whose carrier concentration ranges from semiconductors to degenerate metallic conductors.<sup>2,3</sup> Many studies have been reported for preparation of thin films of CdO.<sup>1,3-10</sup> Recently, we reported on metal organic chemical vapor deposition (MOCVD) of CdO using the cadmium 1,1,1,5,5,5,-hexafluoro-2,4-pentanedionate dihydrate complex,  $Cd(C_5F_6HO_2)_2.2H_2O$ , hereafter  $Cd(hfa)<sub>2</sub>·2H<sub>2</sub>O$ , as a precursor.<sup>2</sup> Moreover, another study

regarding MOCVD of CdO using the volatile  $Cd(hfa)_{2}$ . *N*,*N*,*N*′,*N*′-tetramethylethylenediamine has also been published.8 Both these studies, to date, represent the few examples of MOCVD of CdO.2,8

In this perspective, MOCVD from liquid precursors certainly represents a considerable improvement because of the accurate reproducibility associated with constant evaporation (hence constant mass-transport) rates for given source temperatures.<sup>11</sup> Therefore, in the present communication we report on the novel  $Cd(C_5F_6 HO_2)_2$ <sup> $\cdot$ </sup>CH<sub>3</sub>OCH<sub>2</sub>CH<sub>2</sub>OCH<sub>3</sub> liquid adduct which has proven to be a well-suited precursor for MOCVD of CdO films.

Cadmium-containing compounds are exceedingly toxic and, therefore, care was taken during all sample manipulations.

The  $Cd(C_5F_6HO_2)_2$ <sup> $\cdot$ </sup>CH<sub>3</sub>OCH<sub>2</sub>CH<sub>2</sub>OCH<sub>3</sub> adduct, hereafter  $Cd(hfa)_2$ ·monoglyme, was synthesized by adding, under vigorous stirring, 0.642 g (0.005 mol) of CdO, 1.41 mL (0.01 mol) of H-hfa, and 0.52 mL of monoglyme (0.005 mol) to 40 mL of  $CH_2Cl_2$ . After a few minutes of stirring, the suspension became clear. A colorless oil was obtained after evaporation of the  $CH_2Cl_2$  solvent. A white powder was obtained after the addition of the oil to 30 mL of pentane. Colorless, transparent crystals resulted by dissolving the oil in 90 mL of hexane and leaving the solution to concentrate to room temperature; yield: 99%. Melting point (mp) of the crude product: <sup>72</sup>-73 °C. (Elemental analysis for CdC14H12F12O6. \* To whom correspondence should be addressed. E-mail: agulino@

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**Figure 1.** ORTEP drawing of Cd(hfa)<sub>2</sub>·monoglyme (thermal ellipsoid probability 50%).

Calcd: C, 27.27; H, 1.95. Found: C, 26.94; H, 1.82%). Fast atom bombardment mass spectra (FAB-MS using 3-nitrobenzyl alcohol,  $O_2NC_6H_4CH_2OH = 3NBA$ , as a matrix and cesium as bombarding atoms (35 kV)), (*m*/*z* fragments;  $M = Cd(hfa)_2 \cdot monoglyme$ : 493 (M-CH<sub>3</sub>- $COCHCF<sub>3</sub>$ <sup>+</sup>, 311 (M-monoglyme-COCHCF<sub>3</sub>-CF<sub>3</sub>- $(2F)^+$ , 292 (M-OCH<sub>3</sub>-hfa-F-CF<sub>3</sub>)<sup>+</sup>.<sup>11</sup> The most intense<br>peak at 493 *m*/z(100%) corresponds to the fragment [Cdpeak at 493 *m*/*z* (100%) corresponds to the fragment [Cd-  $(hfa)COCF<sub>3</sub>·OCH<sub>2</sub>CH<sub>2</sub>OCH<sub>3</sub>$ <sup>+</sup>. Infrared transmittance spectra: IR (Nujol; *ν*/cm-1): 3561 (b), 3308 (b), 1647 (s), 1596 (w), 1563 (s), 1530 (s), 1497 (s), 1342 (vw), 1258 (s), 1223 (m), 1191 (m), 1154 (s), 1088 (m), 1060 (m), 1027 (w), 947 (vw), 872 (vw), 853 (m), 830 (w), 797 (s), 768 (w), 740 (m), 722 (m), 670 (s). 1H NMR (CDCl3): *δ* 6.04 (s, 2H of the hfa ligands); 3.72 (s, 4H of methylenic groups of monoglyme), 3.51 (s, 6H of the two methyl groups of monoglyme).11

The solid-state structure of  $Cd(hfa)_2$ ·monoglyme was determined by using single-crystal X-ray diffraction. Experimental conditions and refinement information were similar to those already reported.<sup>12-16</sup> The cadmium cation results as hexacoordinated by the four oxygen atoms of the two hfa ligands and by the two oxygen atoms of the monoglyme molecule (Figure 1). The dihedral angles around the bonds  $O5-C12$ ,  $C12-$ 

**Table 1. Selected Bond Distances (Å) and Angles (deg) of Cd(hfa)2**·**Monoglyme**

$Cd(1)-O(1)$	2.19(1)	$O(1) - Cd(1) - O(4)$	167.2(4)
$Cd(1)-O(4)$	2.20(1)	$O(1) - Cd(1) - O(3)$	92.4(4)
$Cd(1)-O(3)$	2.24(1)	$O(4)-Cd(1)-O(3)$	79.2(4)
$Cd(1)-O(2)$	2.23(1)	$O(1) - Cd(1) - O(2)$	81.8(3)
$Cd(1)-O(5)$	2.34(1)	$O(4)-Cd(1)-O(2)$	93.8(4)
$Cd(1)-O(6)$	2.34(1)	$O(3) - Cd(1) - O(2)$	119.1(4)
		$O(1) - Cd(1) - O(5)$	107.3(4)
		$O(4)-Cd(1)-O(5)$	84.7(4)
		$O(3)-Cd(1)-O(5)$	146.8(3)
		$O(2) - Cd(1) - O(5)$	90.6(4)
		$O(1) - Cd(1) - O(6)$	85.8(4)
		$O(4)-Cd(1)-O(6)$	102.7(4)
		$O(3) - Cd(1) - O(6)$	84.3(4)
		$O(2) - Cd(1) - O(6)$	153.9(4)
		$O(5) - Cd(1) - O(6)$	71.2(4)

**Table 2. Results of the CSD Analysis on the Sketched Fragments**



 $\overline{\phantom{a}}$  = : any kind of bond

C13, and C13-O6 are  $-180(2)$ °,  $-56(2)$ °, and  $179(2)$ °, respectively. These torsion angles imply a transgauche-trans geometric conformation of the monoglyme molecule. The gauche torsional angle around the middle bond of the ether is an obvious consequence of the coordination pattern around the cadmium cation. As expected for the two different ligands, there are two sets of Cd'''O distances (Table 1). Thus, the Cd'''O bonds involving the hfa anions are shorter  $[2.19(1)-2.24(1)$  Å] than those involving the oxygen atoms of the monoglyme molecule [2.34(1) Å]. Related solid-state structures were retrieved from the Cambridge Structural Database  $(CSD)^{17}$  to compare the present  $Cd\cdots O$  distances with those of analogous complexes (see Table 2 sketch **<sup>a</sup>**-**b**) for the searched fragments. Results are summarized as follows:

(i) The  $Cd \cdots O<sub>hfa</sub>$  distances found in Cd(hfa)<sub>2</sub>·monoglyme are in agreement with those found in other cadmium complexes with acetylacetonate (acac) and hfa ligands (Table 2). $2,18$ 

<sup>(12)</sup> Cell parameters and intensity data for  $Cd(hfa)_{2}$ ·monoglyme were obtained on a Nonius CAD4 diffractometer, using graphite monochromated Mo Kα radiation ( $λ = 0.71073$  Å). Intensity data were corrected for Lorentz and polarization effects; an absorption correction was applied once the structure was solved by using the Walker and Stuart method.13 The structure was solved using the SIR-9714 program and subsequently refined by the full-matrix least-squares program SHELX-97.<sup>15</sup> The hydrogen atoms were introduced in calculated positions and their coordinates refined in agreement with those of the linked atoms [the overall temperature factor converged to 0.11(3) Å2]. The cadmium, the fluorine, and the oxygen atoms were refined anisotropically, while the carbon atoms were refined isotropically to improve the data/parameters ratio. Fluorine atoms are in disordered positions and show, as expected, a rather large anisotropic factor.

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<sup>(16)</sup> Cd(hfa)<sub>2</sub> monoglyme crystallizes in the monoclinic system,<br>space group *P*2<sub>1</sub>/*n*, **Z** = 4, cell dimensions *a* = 12.118(7), *b* = 13.066(5),<br> $c = 14.453(7)$  Å,  $\beta = 100.92(7)$ °,  $V = 2247(2)$  Å<sup>3</sup>,  $u = 1.097$  mm<sup>-1</sup>  $c = 14.453(7)$  Å,  $\beta = 100.92(7)$ °,  $V = 2247(2)$  Å<sup>3</sup>,  $\mu = 1.097$  mm<sup>-1</sup>,  $D_c =$ 1.823 g/cm3. The structure was refined to a final *R* factor of 0.0812 based on 2307 independent reflections (339 refined parameters).

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**Figure 2.** DTG (a) and TG (b) of  $Cd(hfa)_2$  monoglyme.

(ii) The  $Cd\cdots O_{\rm monoglyme}$  distances appear shorter than those of Cd complexes with other noncyclic ethers (Table 2).19,20

The Cd center adopts a distorted octahedral geometry with an arrangement of the two hfa ligands similar to that observed in other Cd  $\beta$ -diketonates.<sup>2,18,21</sup> Finally, the present geometry is very similar [RMS value 0.34 Å, calculated as stated in ref 21] to that of the Cd(hfa)<sub>2</sub>.  $2H<sub>2</sub>O$  complex,<sup>2</sup> where the two oxygen atoms of the water molecules are replaced by those of the monoglyme ligand.

The thermal behavior of the  $Cd(hfa)_2$ ·monoglyme was investigated by thermal gravimetric analysis (TGA), under atmospheric prepurified nitrogen, using a 2 °C/ min heating rate in the  $10-400$  °C range.<sup>11</sup> Results show a 93.4% mass loss at 189.2 °C (Figure 2). Moreover, the remaining mass is lost quantitatively in a second step at 330.8 °C. The behavior above 300 °C is probably consistent with some dissociation of the remaining precursor in two fragments:  $Cd(hfa)_2$  and monoglyme.

MOCVD experiments were performed in a horizontal, low-pressure, hot-wall reactor,<sup>2,11</sup> using the present, assynthesized, melted Cd(hfa)<sub>2</sub>·monoglyme adduct as a liquid Cd source. Fused  $SiO<sub>2</sub>$  (quartz) was used as a substrate. Pure Ar  $(70-100 \text{ sccm})$  and  $O_2$   $(100-400 \text{ sccm})$ sccm) were used as carrier and reaction gases, respectively. The total pressure was kept in the 2-6 Torr range. Evaporation rates of the source precursor, suited



Figure 3. UV-visible transmission spectrum for a representative CdO thin film on  $SiO<sub>2</sub>$  substrate obtained using  $400$  sccm  $O<sub>2</sub>$  flow rate.

for MOCVD experiments (4.3 mg/min), were found in the 100-110 °C range. The substrate temperature was maintained in the 320-400 °C range and the deposition time was 120 min.

X-ray diffraction (XRD) measurements of as-deposited films provide evidence of cubic CdO crystallites.<sup>2</sup> Only the (200) reflection ( $\theta = 38.30$ ) was observed, thus pointing to highly textured CdO.2 This behavior was found highly repetitive using the mentioned deposition parameters. The deposited films are yellow and transparent and Figure 3 shows their UV-visible spectrum. The absorption edge is at  $\lambda = 384$  nm (films deposited using  $O_2 = 400$  sccm) and a 90% transmittance value is evident in the visible range. $8,22-27$  The thickness of prototypical films, evaluated from  $UV$ -visible data,<sup>2</sup> is 380 nm ( $O_2$  = 400 sccm) for 120-min experiments. The associated CdO growth rate is 32 Å/min. Similar transmittance and growth rates have already been reported for CdO films.<sup>2,28</sup>

X-ray photoelectron spectra (XPS) were measured with a PHI 5600 Multi Technique System (base pressure of the main chamber  $3 \times 10^{-10}$  Torr).<sup>2,11</sup> The Cd 3d features consist of the main  $3d_{5/2}$  and  $3d_{3/2}$  spin-orbit components at 404.6 and 411.4 eV, respectively.<sup>2,29</sup> It has been reported that atomically clean CdO surfaces,

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in the range 2.438-2.734 Å, while in our complex the two bond distances are 2.34(1) Å.

<sup>(21)</sup> Six of the seven structures, retrieved in the Cambridge Structural Database, show a disposition of the acac derivatives similar<br>to that observed in Cd(hfa)<sub>2</sub>·monoglyme, being the maximum rootto that observed in Cd(hfa)2'monoglyme, being the maximum root-mean-square (RMS) value, calculated using all the atoms in sketch **a** of Table 2, 0.54 Å (for the compound [*cis*-bis(1,3-diphenylpropane-1,3- dionato)dipyridinecadmium(II)] reported in ref 18f).

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exposed a few minutes in air, give the characteristic double XPS O 1s peak, due to the presence of hydroxide species.<sup>2,29</sup> In fact, two O 1s peaks are presently evident at 529.2 and 531.9 eV. In addition, very weak features (at 285.0 and 289.0 eV) due to surface carbon contaminant (2–3 at. %) almost ubiquitous in similarly MOCVD<br>fabricated materials were observed.<sup>2,11</sup> No XPS fluorine signals were detected. Moreover, energy-dispersive Xray (EDX) data point to the absence of any carbon or fluorine in the bulk of the films. Accordingly, XPS depth profiles, obtained eroding CdO films with  $Ar^+$ -ion sputter etching (4 kV, beam current 1.0 *µ*A), do not show evidence of any bulk carbon and fluorine contaminants. Note that, after 1 min of sputtering, only the 529.2 eV O 1s component remains evident because of the removal of surface hydroxide species. Resistivity values of the as-deposited CdO films are  $\leq 6 \times 10^{-4}$  Ω·cm, thus indicating conducting films.<sup>2,25,29</sup> In particular, lower resistivity values were observed for films deposited using low (100 sccm) oxygen flow rates. In CdO the electronic conduction is promoted by the presence of cadmium interstitials or oxygen vacancies which act as n-type defects and produce donor states in the bulk band gap. Therefore, synthesis under more oxidizing condition yields less defective, thus lower conducting, compounds.2,23,26-<sup>28</sup>

In conclusion, a novel  $Cd(hfa)_2$  monoglyme adduct was prepared and characterized by elemental analysis, X-ray single-crystal analysis, mass and NMR spectra, thermal measurements, and IR transmittance spectroscopy. Mild heating (73 °C) resulted in a thermally stable, liquid compound which, in turn, can be easily evaporated. Deposition experiments, in a low-pressure horizontal hot-wall reactor, on SiO<sub>2</sub> substrates, resulted in CdO films. XRD measurements provided evidence that they consist of cubic, (100)-oriented crystals. UVvis spectra showed that the transmittance of asdeposited films in the visible region is about 90%, better than that observed in CdO films obtained using the simple Cd(hfa)<sub>2</sub>·2H<sub>2</sub>O precursor.<sup>2</sup> Finally, resistivity values were lower than those observed for CdO films obtained using the simple  $Cd(hfa)_2.2H_2O$  precursor.<sup>2</sup> The difference between the  $Cd(hfa)_2$ ·monoglyme adduct and that re-ported in ref 8b is due to the different donor ligand attached to the Cd center. Nevertheless, our lowmelting  $Cd(hfa)_2$ ·monoglyme allows MOCVD from liquid precursors, which certainly represents a considerable improvement due to the accurate reproducibility associated with constant evaporation rates for given source temperatures.

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